

Volume-Based Thermodynamics: Estimations for 2:2 Salts

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The lattice energy of an ionic crystal, U_{POT} , can be expressed as a linear function of the inverse cube root of its formula unit volume (i.e., $V_m^{-1/3}$); thus, $U_{POT} \approx 2I(\alpha/V_m^{1/3} + \beta)$, where α and β are fitted constants and I is the readily calculated ionic strength factor of the lattice. The standard entropy, S, is a linear function of V_m itself: $S \approx kV_m + c$, with fitted constants k and c. The constants α and β have previously been evaluated for salts with charge ratios of 1:1, 1:2, and 2:1 and for the general case q:p, while values of k and c applicable to ionic solids generally have earlier been reported. In this paper, we obtain α and β , k and c, specifically for 2:2 salts (by studying the ionic oxides, sulfates, and carbonates), finding that U_{POT} {MX 2:2}/(kJ mol⁻¹) \approx 8(119/ $V_m^{1/3}$ + 60) and S° {MX 2:2}/(J K⁻¹ mol⁻¹) \approx 1382 V_m + 16.

Introduction

Volume-based thermodynamics¹ (VBT) provides a means of estimating thermodynamic data using the formula unit volume of materials, $V_{\rm m}$, which is easily and readily determined directly from crystal structure data ($V_{\rm m} = V_{\rm cell}/Z$, where Z is the number of formula units in the unit cell, of volume $V_{\rm cell}$) or from density or by ion volume additivity ($V_{\rm m} = V_+ + V_-$), using the individual ion volumes found in our databases.^{2a-c} The VBT approach (as well as the use of *volume* as a criterion of the ion size) is gaining wide acceptance,^{2d} finding application in tackling diverse topical problems. The link to earlier work on 1:1 charged salts by Mallouk et al.^{2e} has been referred to elsewhere.^{2a}

The constants, α and β (whose values vary slightly with the stoichiometry of the lattice and with the set of materials used in the statistical fitting), of the lattice potential energy equation (1) have previously been evaluated³ for salts with charge ratios of 1:1, 1:2, and 2:1 and for the general case M_pX_q , with a charge ratio *q*:*p*, but not specifically for the important case of ionic lattices for the 2:2 case, of salts possessing both a divalent cation and anion:

$$U_{\rm POT} \approx 2I(\alpha/V_{\rm m}^{-1/3} + \beta) \tag{1}$$

The ionic strength factor,⁴ I, in this equation is defined by the equation

$$I = \frac{1}{2} \sum_{\text{ions}}^{\text{formula}} n_i z_i^2$$
(2)

where n_i is the number of each type of ion, of integer charge z_i , in the formula unit (with complex ions such as sulfate being counted as a single, doubly charged ion^{5,6}), yielding

(3) Jenkins, H. D. B.; Tudela, D.; Glasser, L. Inorg. Chem. 2002, 41, 2364.

(5) Note that complex ions can equally well be regarded as decomposing into separated single ions, such as SO₄²⁻ yielding S₆⁺ plus O²⁻; this will result in a different, and much larger, value of the lattice energy, which includes the self-energy of the complex ion. For a more detailed outline, see, for example: Jenkins, H. D. B.; Waddington, T. C. J. Chem. Phys. **1972**, 56, 5323. For lattice energies greater than 5000 kJ mol⁻¹, the linear equation (1) does not apply and, instead, a generalized equation⁶ should be used. Conventionally, however, recognized complex ions, such as sulfate, are often treated as single entities although this may not apply when dealing with more complex materials such as in silicate minerals.

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⁽¹⁾ Glasser, L.; Jenkins, H. D. B. Chem. Soc. Rev. 2005, 10, 866.

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⁽⁴⁾ Glasser, L. Inorg. Chem, 1995, 34 (20), 4935.

⁽⁶⁾ Glasser, L.; Jenkins, H. D. B. J. Am. Chem. Soc. 2000, 122, 632.

Volume-Based Thermodynamics Estimations for 2:2 Salts

the value I = 4 for 2:2 salts. Similarly, the values of standard entropies, S° , of ionic solids are observed to be rather closely linearly related to the formula unit volume,⁷ as in eq 3. These

$$S^{\circ} \approx kV_{\rm m} + c$$
 (3)

empirical constants are here determined for 2:2 salts by plotting the lattice potential energies, $U_{\rm POT}$ (as determined from Born–Fajans–Haber thermochemical cycles⁸), against $V_{\rm m}^{-1/3}$ of known ionic salts (by studying the ionic oxides, sulfates, and carbonates). In general, it is recommended that, whenever possible, such plots be made by employing compounds as closely generic as possible to the target compound under study. The resulting predictions will then, inevitably, be more reliable.

Equations 1 and 3 form the backbone of VBT, which has considerably extended the range of materials for which thermodynamics can be estimated. The key factor giving this approach its range of application and simplicity when compared to other methods is the ease by which formula unit volumes may be assigned to hypothetical as well as known salts. It can therefore be adopted and used with ease by nonspecialists, requiring only a calculator. The VBT approach challenges presently used approaches based on complex theoretical calculations, while it can often be used effectively in tandem with them^{1,2d} but does assume that the materials are fully ionic; procedures are currently in development to deal with materials that have a substantial component of covalency.

Calculations for $U_{\rm POT}$ and S°

Materials have been selected for the current exercise as those being generally regarded as having a high degree of ionicity, that is, oxides, sulfates, and carbonates of the divalent metallic elements; formula unit volumes have been obtained from crystallographic sources⁹ as being most reliable. Standard entropies have been collected from the Handbook of Chemistry and Physics,¹⁰ and lattice energies have been calculated, in a Born-Haber-Fajans thermochemical cycle, by summing the standard enthalpies of formation, $\Delta_f H^\circ$, of the relevant ions¹¹ (also see the table in the Supporting Information) and subtracting the standard enthalpy of formation of the corresponding ionic solid.¹⁰ While $\Delta_f H^\circ$ for the cations is well-established (being the sum of the sublimation enthalpy of the solid metal plus the first and second ionization potentials of the gaseous species),¹¹ the corresponding formation enthalpies of the doubly charged anions are unknown because they do not have bound gaseous states. Instead, for the anions, values are determined by backcalculation of the thermochemical cycles for 1:2 compounds of known formation enthalpy (say, Na₂SO₄); the



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- (11) (a) Marcus, Y. *Ion Properties*; Marcel Dekker: New York, 1997. (b) NIST database: http://www.webbook.nist.

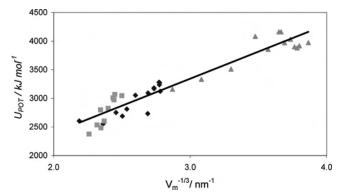


Figure 1. Plot of $U_{\rm POT}$ versus $V_{\rm m}^{-1/3}$ for the 2:2 salts in Table 1: diamonds for carbonates, squares for sulfates, and triangles for oxides. Linear least-squares-fitted equation: $U_{\rm POT}$ {MX 2:2}/(kJ mol⁻¹) $\approx 8(119/V_{\rm m}^{1/3} + 60)$ [correlation coefficient $R^2 = 0.91$ and degrees of freedom = (no. of points -2) = 33, leading to the statistically based conclusion that there is greater than 99.9% probability that the assumption of linearity between $U_{\rm POT}$ and $V_{\rm m}^{-1/3}$ is correct].

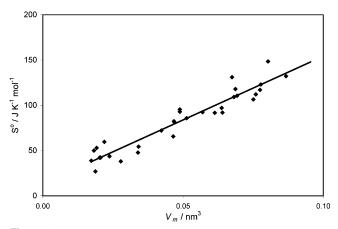


Figure 2. Plot of S° versus $V_{\rm m}$ for the 2:2 salts in Table 1. Linear leastsquares-fitted equation: S° {MX 2:2}/(J K⁻¹ mol⁻¹) $\approx 1412V_{\rm m} + 14$ [correlation coefficient $R^2 = 0.90$ and degrees of freedom = (no. of points -2) = 28, leading to the statistically based conclusion that there is greater than 99.9% probability that the assumption of linearity between S° and $V_{\rm m}$ is correct]. See also the Note Added in Proof.

resulting average values are assumed to apply equally to the 2:2 compounds.

The least-squares linear fit of the plot of U_{POT} versus the crystallographic-based $V_{\text{m}}^{-1/3}$ (Figure 1) takes the form

$$U_{\rm POT}$$
{MX 2:2}/(kJ mol⁻¹) $\approx 8(119/V_{\rm m}^{1/3} + 60)$ (4)

The mean absolute error of the fit is only 4.4%, with the largest error being 11.2% for CuCO₃, which may be expected to have significant covalency. The values of α and β in eq 4 (119 and 60, respectively) may be compared with the original estimations (102 and 95, respectively).

A corresponding plot for standard entropies (S° versus $V_{\rm m}$) appears in Figure 2. The least-squares linear fit of the plot of S° versus the crystallographic-based $V_{\rm m}$ for the ionic solids⁷ in Table 1 (taking into account asterisked data, see Note Added in Proof) (Figure 2) takes the form

$$S^{\circ}{MX 2:2}/(J K^{-1} mol^{-1}) \approx 1382V_{m} + 16$$
 (5)

The mean absolute error of the fit is 7.7 J K^{-1} mol⁻¹, while

Table 1. $\Delta_{\rm f} H^{\circ}$, $V_{\rm m}$, $U_{\rm POT}$, and S° Values for 2:2 Ionic Salts

| | | | | | S°/ |
|------------------|-------------------|--|-----------------|-----------------|--------------------|
| | | $\Delta_{\rm f} H^{\circ} / V_{\rm m} /$ | | $U_{\rm POT}$ | (J K ⁻¹ |
| mineral | formula | $(kJ mol^{-1})$ | nm ³ | $(kJ mol^{-1})$ | mol^{-1}) |
| rhodochrosite | MnCO ₃ | -894.1 | 0.0513 | 3093 | 85.8 |
| siderite | FeCO ₃ | -740.6 | 0.0489 | 3170 | 92.9 |
| sphaerocobaltite | CoCO ₃ | -713 | 0.0468 | 3236 | 88.7* |
| - | $ZnCO_3$ | -812.8 | 0.0468 | 3275 | 82.4 |
| otavite | $CdCO_3$ | -750.6 | 0.0570 | 3053 | 92.5 |
| cerussite | PbCO ₃ | -699.1 | 0.0675 | 2751 | 131 |
| smithsonite | ZnCO ₃ | -818.9 | 0.0469 | 3281 | 81.2 |
| siderite | FeCO ₃ | -752 | 0.0489 | 3181 | 95.47 |
| rutherfordine | UO_2CO_3 | -1716.4 | 0.0955 | 2605 | |
| | CuCO ₃ | -596.2 | 0.0514 | 2733 | 87.9* |
| | MgCO ₃ | -1095.8 | 0.04651 | 3123 | 65.7 |
| | CaCO ₃ | -1207.6 | 0.06131 | 2813 | 91.7 |
| | SrCO ₃ | -1220.1 | 0.06378 | 2690 | 97.1 |
| | $BaCO_3$ | -1213 | 0.07595 | 2552 | 112.1 |
| | $BaSO_4$ | -1473.2 | 0.0867 | 2376 | 132.2 |
| | $CaSO_4$ | -1434.5 | 0.0751 | 2602 | 106.5 |
| | $SrSO_4$ | -1453.1 | 0.0775 | 2486 | 117 |
| | $PbSO_4$ | -920 | 0.0803 | 2535 | 148.5 |
| | $MnSO_4$ | -1065.3 | 0.0724 | 2827 | 122.2* |
| | $CoSO_4$ | -888.3 | 0.0687 | 2975 | 118 |
| | ZnSO ₄ | -982.8 | 0.0693 | 3008 | 110.5 |
| | $NiSO_4$ | -872.9 | 0.0641 | 3046 | 92 |
| | $CuSO_4$ | -771.4 | 0.0682 | 3067 | 109.2 |
| | $CdSO_4$ | -933.3 | 0.0778 | 2799 | 123 |
| | MgO | -601.6 | 0.01875 | 3900 | 27 |
| | CaO | -634.9 | 0.02783 | 3511 | 38.1 |
| | SrO | -592 | 0.03415 | 3333 | 54.4 |
| | BaO | -548 | 0.04228 | 3158 | 72.1 |
| | TiO | -519.7 | 0.01822 | 3918 | 50 |
| | VO | -431.8 | 0.0173 | 3973 | 38.9 |
| tenorite | CuO | -157.3 | 0.0203 | 4161 | 42.3 |
| manganosite | MnO | -385.2 | 0.0220 | 3855 | 59.7 |
| wustite | FeO | -272 | 0.0199 | 3972 | 50.8* |
| | CoO | -237.9 | 0.0192 | 4032 | 53 |
| zincite | ZnO | -350.5 | 0.0238 | 4083 | 43.7 |

the maximum error of 21.8 J K⁻¹ mol⁻¹ is for PbCO₃. The values of *k* and *c* in eq 5 (1382 and 16, respectively) may be compared with the estimations for all ionic solids⁷ (1360 and 15, respectively).

Table 2. Updated Coefficients α and β for Various Stoichiometries in Equation 1

| - | | | | | |
|-------------------------------------|--------------------------|-----------------------------|-----------------------|------|--|
| salt (charge ratio) ^a | ionic strength factor, I | $\alpha/(kJ mol^{-1} nm)^b$ | $\beta/(kJ mol^{-1})$ | | notes |
| MX (1:1) | 1 | 117 | 52 | 0.94 | |
| $MX_2 (2:1)^a$ | 3 | 134 | 61 | 0.83 | n = 13 |
| $M_2X (1:2)^a$ | 3 | 165 | -30 | 0.95 | n = 21, plot shown in Figure 1^{2a} |
| MX (2:2) | 4 | 102 | 92 | | as originally quoted3 |
| | | 119 | 60 | 0.91 | values in this study $(n = 35)$ |
| $M_p X_q (q:p)^a$ general salts | $^{1/2}(pq^{2}+qp^{2})$ | 139 | 28 | 0.91 | . , |

^{*a*} The ratios have been revised to represent *charge* ratios and are different from the *ion* ratios used in the original paper.^{2a} ^{*b*} As always, the magnitude of $\alpha/(kJ \text{ mol}^{-1} \text{ nm})$ is similar to that of the electrostatic factor, $A/(kJ \text{ mol}^{-1}) = 121.4$ but is fitted to the data available.

Discussion

By establishing the constants for the lattice potential energy and standard entropy equations for handling 2:2 salts, we have now completed the suite of equations³ that is commonly needed for the application of the VBT approach (collected in Table 2).

Note Added in Proof: The asterisked $S^{\circ}/(J \text{ K}^{-1} \text{ mol}^{-1})$ values in Table 1 were not included in the original data set, but, if added, they lead to a modified fit, $S^{\circ}\{MX 2:2\}/(J \text{ K}^{-1} \text{ mol}^{-1} \approx 1381.7V_{\text{m}} + 16.0$, having $R^2 = 0.90$ and n = 34; this modified fit value is actually closer to our previously quoted⁷ value for general ionic solids.

Supporting Information Available: Table of enthalpies of formation of gaseous ions. This material is available free of charge via the Internet at http://pubs.acs.org.

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